

Development of 3D graphene structures and their prospective applications

M. Christian^{1,2}, L. Venturi^{2,3}, L. Ortolani¹, R. Rizzoli¹, Z. Xia², V. Palermo², V. Margaria⁴, A. Sacco⁴, N. Garino⁴, A. Chiodoni⁴, M. Quaglio⁴, C.F. Pirri^{4,5} and V. Morandi¹

¹IMM, CNR, Bologna, Italy ²ISOF, CNR, Bologna, Italy ³University of Bologna Department of Physics, Bologna, Italy ⁴Center for Space Human Robotics @Polito, Istituto Italiano di Tecnologia, Corso Trento 21, 10129 Torino, Italy ⁵Applied Science and Technology Department, Politecnico di Torino, Corso Duca degli Abruzzi 24, 10129 Torino, Italy

CIMM Institute for Microelectronics and Microsystems **Bologna** Section











Ni foam Ni-G C NI-G-PMMA [2] CH4/H2/Ar **PMMA** 1,000 °C coating FeCl₃/HCl Ni etching PDMS Acetone PMMA infiltration dissolving GF/PDMS GF GF-PMMA

Applications

 Electrodes in supercapacitors, wearable devices, dyesensitised solar cells, Li-ion batteries

3D graphene materials

e.g. graphene foams

- Electrochemical catalysts
- Gas sensing
- Adsorbents
- Thermal interface material
- Composites with polymers for flexible, foldable and stretchable conductors

CuO nanoflowers



Biosensing

Ni(OH)₂



H₂ fuel cells

Superhydrophobic coatings

Teflon

1. J Mater Chem C 2 (2014) 1879-1886 2. Nat Mater 10 (2011) 424-428

- 3. Biosens Bioelectron 59 (2014) 384-388
- 4. Int J Hydrogen Energy 39 (2014) 7876-7884

5. Small 9 (2013) 75-80

[5]



Metal foam precursors are washed thoroughly by ultrasonication in dilute HCl, DI water and acetone to remove contaminants

- Templates are annealed at 1000 ^oC under 50 sccm H₂ for 30 min to remove surface oxides
- Deposition is performed at atmospheric pressure under 500 sccm H₂ and 50 sccm CH₄ for 10 min

Ni is etched in 4.5% FeCl₃ at 80 °C overnight then salt residuals are removed using 10% HCl at 80 °C





Graphene foam



σ ~ 10 S/cm



GF retains the structure of the Ni foam after it has been etched away, even without polymer support. Pore size ~100-400 μm. Hollow interior with graphene wall thickness 10-20 nm.



500 sccm H_2 + 100 sccm CH_4



Could not be collected after PMMA removal





Collapsed into a film



100 sccm H_2 + 100 sccm CH_4

Free-standing but broken graphene foam

The different appearance using Cu is most likely a result of the surface adsorption growth mechanism of graphene on Cu compared to dissolution-segregation-precipitation on Ni [6]



A smaller pore size could result in GF with better structural integrity as well as a greater volumetric energy density

GF were synthesised on a template of Ni nanoparticles, sintered into a 3D network



Resulting GF had few layer and multilayer areas, few defects and much smaller pores BUT the pore size distribution was not well controlled

To try to control the pore size distribution, nanoparticles were first pressed into pellets but the minimum pressure reachable using a hydraulic press was too high to form porous materials







Optimised GF were grown on a template of a mixture of Ni and NiO nanoparticles, handpressed into a pellet with ~50 kg/cm² pressure



As before, the resulting GF had few layer and multilayer areas, and few defects but also a well-controlled pore-size distribution in the range of <1-5 μ m



Previous work in our groups:







Zhen Yuan Xia ^a, Di Wei ^b, Elzbieta Anitowska ^a, Vittorio Bellani ^c, Luca Ortolani ^d, Vittorio Morandi ^d, Massimo Gazzano ^a, Alberto Zanelli ^a, Stefano Borini ^b, Vincenzo Palermo ^{a,*}

CrossMark

^a Istituto per la Sintesi Organica e la Fotoreattività – Consiglio Nazionale delle Ricerche, via Gobetti 101, 40129 Bologna, Italy ^b Nokia R&D UK Ltd, Broers Building, 21 J. J. Thomson Avenue, CB3 0FA Cambridge, UK ^c Dipartimento di Fisica and CNISM, Universitá degli Studi di Pavia, via Bassi 6, 27100 Pavia, Italy ^d Istituto per la Microelettronica e Microsistemi – Consiglio Nazionale delle Ricerche, via Gobetti 101, 40129 Bologna, Italy

Combining GF with Fe_2O_3 leads to significantly improved specific capacity. A free-standing GF/Fe₂O₃ composite could be used as an electrode in Li-storage materials without the need for a Cu or Ni current collector or polymer binder.



The cycling capability of the GF was poor, perhaps due to the discontinuous nature of the EGO. Could CVD GF provide a better result?



A first test of the electrochemical properties was performed on GF with Ni still inside:



Conditions: 3-electrode configuration, with working electrode CVD graphene on Ni foam (\emptyset 9 mm) and lithium foil as counter/reference electrode. Electrolyte 1 M LiPF₆ (LP30, BASF) in a 1:1 (w/w) mixture of ethylene carbonate (EC) and dimethyl carbonate (DMC)

Electrochemistry of Materials for Energetics



Given these promising results, measurements were also performed on free-standing GF:



The GF showed excellent performance over 100 cycles. However, they were very fragile so a lot of care needed to be taken when assembling the cells!





CVD GF were functionalised by a chemical procedure similar to that used for the EGO GF



Due to the hydrophobicity of CVD graphene, the coating was sparse and uneven



To create a more even coating, the deposition was instead performed electrochemically



Porous Iron Oxide was deposited at an anodic current density of 0.125 mA cm⁻² in an aqueous solution of 0.2 M CH_3COONa , 0.1 M Na_2SO_4 and 0.1 M $Fe(NH_4)_2(SO_4)_2.6H_2O$ at room temperature for 15 minutes, and further annealing at 300 °C for 1 hour [7]



In this way, the outer branches of GF were evenly decorated with a layer of flower-like Fe_2O_3 nanocrystals. However, the inner branches were not completely covered



A series of treatments were attempted to improve coverage of the inner branches of GF



Ineffective



not easily reproducible

Effective as EtOH wets hydrophobic GF

Plasma pre-treatment



0.022

0.02

100

2V, 5 min

Time (s)

Effective oxidation of all surfaces

Electrochemical performance to be confirmed!

Reduced pore-size GF are less fragile than those with very large empty pores. Preliminary measurements showed good potential but some deterioration, probably due to





These GF were functionalised with Fe_2O_3 nanocrystals as before. They were evenly decorated with nanocrystals, even without pre-treatment. Electrochemical measurements are to be performed





Microbial fuel cells have the ability to convert wastewater and other organic matter into electricity using the natural processes of microorganisms







GF is used as a charge collector and a catalyst for the ORR. Even without modification it performs much better than expected for a simple graphene film – probably residual Fe nanoparticles from the etching procedure provide additional catalytic activity.





GF@anode



GF is used as a substrate for the biofilm, showing significant improvement over standard carbon felt. However, the improvement should be much greater for this class of materials – once again the fragility of the GF leads to deterioration in the result over time.





- High quality GF were simply produced by CVD and the template etching process optimised
- GF with reduced and controlled pore size (1-10 μ m) were produced using Ni/NiO nanoparticle templates. These have the potential to overcome problems of fragility while simultaneously introducing new possibilities for applications
- GF of both kinds show good potential for use as battery electrodes. Compared to previous tests, there was little or no deterioration of the capacity over 100 cycles
- CVD GF are very hydrophobic, making them difficult to functionalise by chemical means but after suitable pre-treatments (e.g. ethanol or plasma), CVD GF were successfully functionalised with flower-like Fe₂O₃ nanocrystals using electrochemical methods. Electrochemical measurements are in progress
- GF showed great potential for use as both anode and cathode materials for microbial fuel cells, but problems of fragility need to be overcome



Thank you for your attention

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- Facile synthesis of Fe(OH)₃ colloidal by hydrolysis of FeCl₃
- Avoid hydrothermal routes with high pressure and long time
- Fe(OH)₃ could be transformed to α -Fe₂O₃ nano-crystalline by calcination



Xia et al., Carbon **2015***, 84,* 254-262



Crystallites of Fe₂O₃ are estimated to be 5 nm from Scherrer equation





Electrochemical measurements

The working electrodes was CVD graphene on Ni foam (\emptyset 9 mm). The electrodes were measured in an electrolyte-filled jar, using a three-electrode configuration with lithium foil as counter/reference electrode. The electrolyte was 1 M LiPF₆ (LP30, BASF) in a 1:1 (w/w) mixture of ethylene carbonate (EC) and dimethyl carbonate (DMC). Cell assembly was performed in Ar atmosphere MBraun Labmaster SP dry box (H₂O and O₂ < 0.1 ppm) and all the electrochemical measurements were carried out by Biologic VSP and PerkineElmer VMP potentiostats/galvanostats.

